

Light-Emitting Electrochemical Cells and Solution-Processed Organic Light-Emitting Diodes Using Small Molecule Organic Thermally Activated Delayed Fluorescence Emitters

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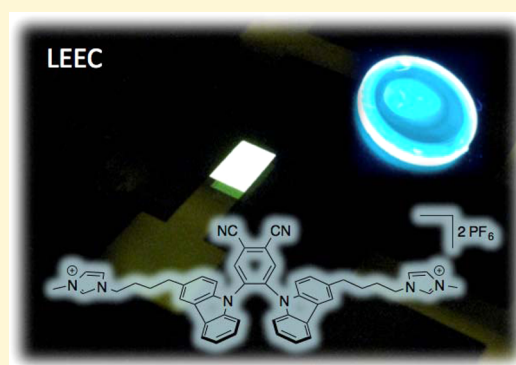
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S Supporting Information

ABSTRACT: Two novel charged organic thermally activated delayed fluorescence (TADF) emitters, **1** and **2**, have been synthesized. Their TADF behavior is well-supported by the multiexponential decay of their emission (nanosecond and microsecond components) and the oxygen dependence of the photoluminescence quantum yields. Spin-coated electroluminescent devices have been fabricated to make light-emitting electrochemical cells (LEECs) and organic light-emitting diodes (OLEDs). The first example of a non-doped charged small organic molecule LEEC is reported and exhibited an external quantum efficiency (EQE) of 0.39% using **2**. With a multilayer architecture, a solution-processed OLED device using neat **2** as the emitting layer gave an EQE of 5.1%, the highest reported to date for a nondoped solution-processed small molecule organic TADF OLED. These promising results open up a new area in light-emitting materials for the development of low-cost TADF LEECs.



INTRODUCTION

Organic light-emitting diodes (OLEDs) have come to the fore as state-of-the-art technology for visual displays and lighting.^{1–4} OLEDs are desirable as they are lightweight and flexible, provide better contrast, and possess a large viewing angle.⁴ OLEDs are also more power efficient than traditional lighting sources, and thus, their wide adoption can alleviate significantly the strain on current energy demand because lighting alone constitutes ~20% of electricity consumption worldwide.⁵ The “first-generation” OLEDs were based on organic fluorescent emitters whose efficiency was intrinsically capped at 25% due to only being able to recruit singlet excitons.⁶ The “second-generation” OLEDs employed organometallic phosphorescent emitters, which harvest both singlet and triplet excitons for emission due to the enhanced intersystem crossing (ISC) mediated by the large spin–orbit coupling of the heavy metals such as iridium(III) and platinum(II).^{7–9} Despite their highly desirable performance characteristics, the rarity of these metals, their high cost, and their toxicity are important detracting features that inhibit large-scale, worldwide adoption of OLED technology, particularly for lighting applications. “Third-generation” OLEDs were recently reported by Adachi and co-workers. His group demonstrated how small organic molecules, emitting via a thermally activated delayed fluorescence (TADF) mechanism,¹⁰ could be integrated into OLEDs and exhibit very

high efficiencies as, like with phosphorescent emitters, both singlet and triplet excitons are recruited for emission.^{11–14} Thus, TADF-based OLEDs address the key detracting features endemic to second-generation OLEDs while retaining their advantages.¹⁵

The principle of TADF relies on a small energy gap between the singlet and triplet excited states (ΔE_{ST}). Under these conditions, the excitons in the triplet state can return to the singlet state by reverse intersystem crossing (RISC) using thermal energy, followed by radiative fluorescence.¹¹ The small ΔE_{ST} is realized by spatial separation between HOMO and LUMO to minimize the exchange integrals between the two orbitals.¹⁶ A large number of organic TADF emitters have been reported to date, and the majority of them are based on a twisted intramolecular charge transfer (TICT) design in which the donor and acceptor moieties are designed to be nearly orthogonal to each other.^{11,13,17,18}

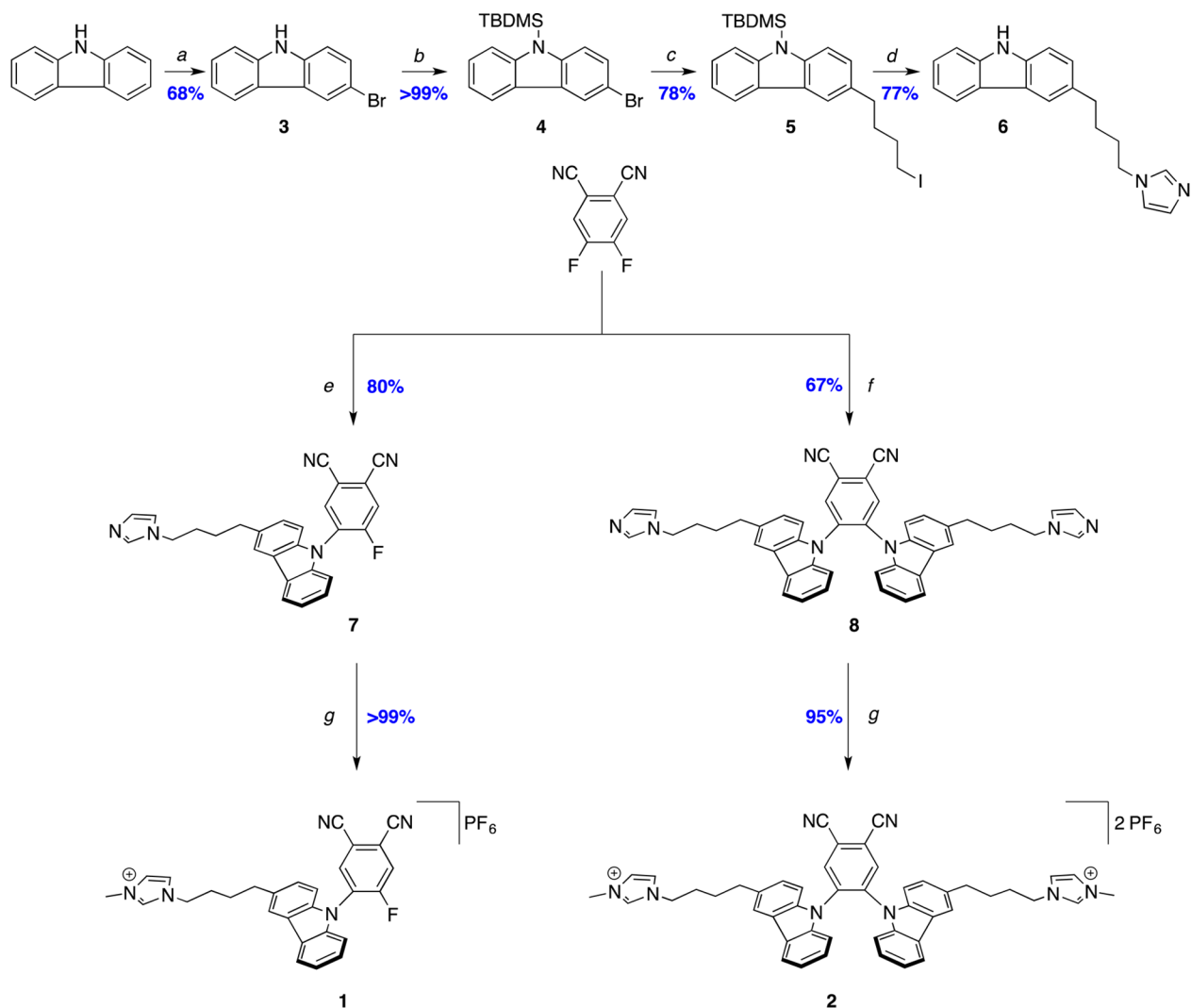
Despite this important paradigm shift in emitter design, current OLEDs, including TADF OLEDs, still employ air sensitive electrodes requiring rigorous encapsulation, are usually vacuum deposited, and possess a complex multilayer

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Scheme 1. Synthesis of the Charged TADF Emitters for LEECs **1** and **2**^a

^aConditions: (a) NBS, MeCN, room temperature, 1 h; (b) (i) NaH, THF, room temperature, 30 min, (ii) TBDMSCl; (c) (i) *n*-BuLi, THF, -78°C , 30 min, (ii) excess 1,4-diiodobutane; (d) NaH, imidazole, THF, reflux, 4 h; (e) (i) NaH, **6**, THF, room temperature, 30 min, (ii) 2 equiv of 4,5-difluorophthalonitrile, room temperature, 4 h; (f) (i) NaH, **6**, THF, room temperature, 30 min, (ii) 0.6 equiv of 4,5-difluorophthalonitrile, room temperature, 4 h; (g) (i) MeI, MeCN, 40°C , 2 h, (ii) saturated NH_4PF_6 (aq) (NBS, *N*-bromosuccinimide; TBDMSCl, *tert*-butyldimethylsilyl chloride).

architecture that adds to the cost of fabrication. Single-layer solid-state light-emitting electrochemical cells (LEECs) have received much recent attention for their potential to address these negative features of OLEDs.^{19–22} The emitters in LEECs are frequently ionic transition metal complexes (iTMCs), the most popular and highest-performing class of which are cationic iridium(III) complexes.^{23,24} Without a similar paradigm shift in emitter design coupled with improved device performance characteristics, LEECs will suffer a worse fate than second-generation OLEDs, and this promising technology will be relegated to academic inquiry. Recently, Bolink and co-workers demonstrated the first example of an operational LEEC with a small molecule charged organic cyanine-based fluorophore,²⁵ while Edman showed how a small molecule neutral organic fluorophore can be used as a dopant in LEECs.²⁶ We now report the first example of an organic TADF-based LEEC wherein the TADF material serves the dual role of charge transporter and emitter. Additionally, we also report doped and single-component spin-coated OLEDs, whose external quan-

tum efficiencies are superior to those few previous reports of nondoped TADF OLEDs in the literature,^{17,27,28} and contrast their performance to that of the LEEC.

Small molecule emitters must be charged to mediate charge transport in a single-layer LEEC. We thus set about incorporating tethered imidazolium units onto an *N*-carbazoyl-dicyanobenzene scaffold first reported by Adachi¹¹ to build charged analogues, **1** and **2**, whose synthesis is illustrated in Scheme 1.

The preparation of 3-bromocarbazole **3** using *N*-bromosuccinimide²⁹ was contaminated by some starting material and 3,6-dibromocarbazole, both of which could be removed by fractional recrystallization from toluene. The preparation of **5** was accomplished in good yield by dropwise introduction of the lithiated TBDMS-protected 3-bromocarbazole **4** intermediate to excess 1,4-diiodobutane. Compound **6** was obtained by $\text{S}_{\text{N}}2$ reaction of sodium imidazolite with **5** followed by silyl deprotection using sodium hydride in a one-pot fashion.³⁰ Compounds **7** and **8** were synthesized by reacting **6** with

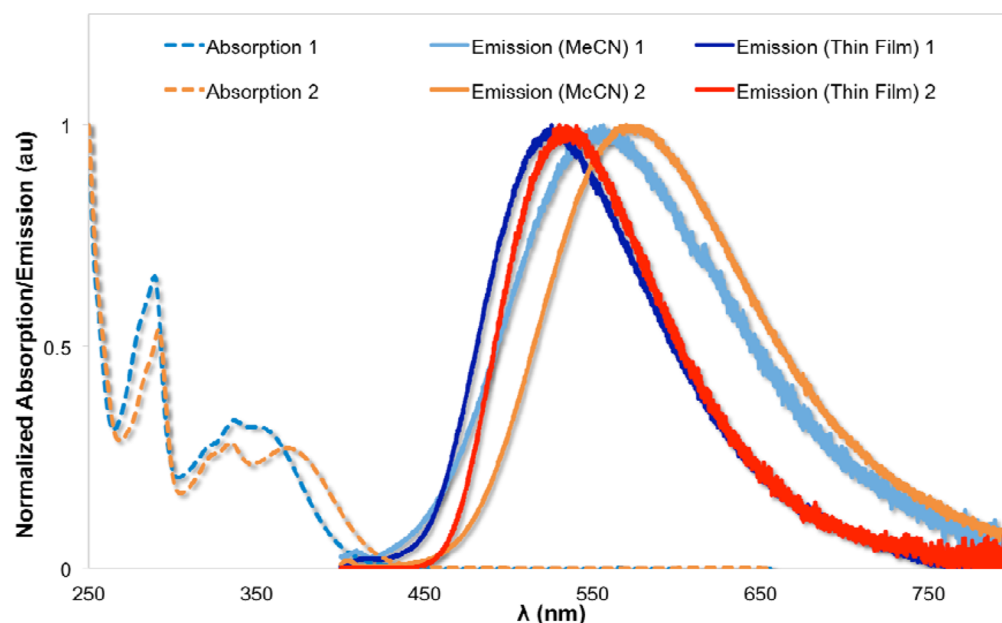


Figure 1. Normalized absorption spectra of **1** and **2** in aerated MeCN at 298 K and normalized emission spectra in deaerated MeCN at 298 K and in pristine thin film under air.

substoichiometric and stoichiometric 4,5-difluorophthalonitrile, respectively, under basic conditions via nucleophilic aromatic substitution. The targeted charged TADF emitters **1** and **2** were obtained following methylation with iodomethane and anion metathesis with a saturated aqueous NH_4PF_6 solution in 33 and 27% yields, respectively, over six steps. The solubilities of **1** and **2** in DCM were greatly improved after the anion metathesis.

Both **1** and **2** exhibit irreversible oxidation and reversible oxidation waves in MeCN solution by cyclic voltammetry (Table S1 and Figure S20). The HOMO of **1** (−5.93 eV) is slightly lower than that of **2** (−5.87 eV) because of the presence of the electron-withdrawing fluorine atom. The LUMO of **1** (−2.92 eV) is slightly higher than that of **2** (−2.99 eV), which is due to the increased level of conjugation imparted by the second carbazole moiety that lowers the LUMO in **2**. These structure–property relationships are mirrored in the absorption spectra wherein the absorption profile of **2** is slightly red-shifted compared with that of **1** (Figure 1).

Solution photoluminescence of the two materials was investigated in MeCN. The primary photophysical parameters in solution are detailed in Table 1 (absorption data reported in Table S2).

The emission in MeCN solution and in film for both **1** and **2** (Figure 1) is broad and unstructured, which is characteristic of CT emission. The excitation and absorption spectra match, corroborating the high level of purity determined by the compound characterization. The solution-state emission for **1** and **2** is red-shifted by ~30 and ~40 nm, respectively,

compared to emission in film (*vide infra*). In solution, the emission for both **1** and **2** is weak and the observed emission lifetimes are in the nanosecond regime. There is little change in the photophysical properties upon removal of O_2 , suggesting that in MeCN **1** and **2** act as fluorophores with no observed TADF. This result, coupled with the observation that PL is red-shifted in solution compared to thin film, indicates that solvent interactions may lower the energy of the triplet (T_1) with respect to the singlet (S_1), resulting in an increased ΔE_{ST} , and thus, TADF would be switched off. By point of comparison, dicarbazolyldicyanobenzene (2CzPN), which does not contain the tethered imidazolium units, was shown by Adachi to emit at 473 nm in a toluene solution via TADF with a photoluminescence quantum yield, Φ_{PL} , of 47%.¹¹

Photoluminescence in thin films is the process of interest when considering these materials as TADF emitters. In a spin-coated neat film, **1** and **2** show significantly stronger emission than in solution, with Φ_{PL} values of 0.093 and 0.207, respectively, under aerated conditions. As interactions between chromophores often lead to quenching of PL, we also investigated the emission when the materials were diluted in a PMMA host. In samples with 10 wt % **1** and **2** in PMMA, Φ_{PL} improved substantially with Φ_{PL} values of 0.447 and 0.730 for **1** and **2**, respectively. Table 2 provides full details of the measured photophysical properties of the films.

The long (microsecond) average lifetime and the shortening of the lifetime in air suggest a triplet-derived TADF process is occurring, in contrast to the solution data. The PL decay is observed to be complex, spanning 4 orders of magnitude of time and requiring 7–9 exponential functions to adequately describe it. These exponentials are, however, simply parametrization as the decay is derived from a mixture of initial fluorescence from the singlet (often denoted as prompt fluorescence), and delayed fluorescence mediated by intersystem crossing (ISC) followed by slow reverse intersystem crossing (RISC) populated emission from the singlet, subject to Boltzmann statistics. Consequently, to permit a fair comparison of the two materials, and between aerated and vacuum

Table 1. Photoluminescence of **1** and **2** in MeCN at 298 K

	λ_{em} (nm) ^{a,b}	λ_{em} (nm) ^{a,c}	Φ_{PL} ^{b,d}	Φ_{PL} ^{c,d}	τ_e (ns) ^{b,e}	τ_e (ns) ^{c,e}
1	556	558	0.014	0.029	2.90	3.26
2	574	572	0.021	0.026	4.44	4.97

^a λ_{exc} at 360 nm. ^bMeasured in aerated solvent. ^cMeasured in degassed solvent. ^dQuinine sulfate used as the reference ($\Phi_{\text{PL}} = 0.546$ in 1 N H_2SO_4 at 298 K). ^e λ_{exc} at 375 nm.

Table 2. Photoluminescence of **1** and **2** in Neat and Doped Thin Films

		film (neat)			film (10% doped in PMMA)		
		λ_{em} (nm)	Φ_{PL}^a	τ_{ave} (μs)	λ_{em} (nm)	Φ_{PL}^a	τ_{ave} (μs)
1	aerated	530	0.09	0.26	490	0.44	2.54
	degassed		0.10	0.96		0.56	6.14
2	aerated	536	0.21	0.97	500	0.73	1.08
	degassed		0.21	2.73		0.90	6.34

^aPLQY measured using a Hamamatsu Absolute PL quantum yield measurement system, with an integrating sphere and nitrogen purge where appropriate.

conditions, an average lifetime can be calculated that is the weighted average of the individual components with their pre-exponential factors. Full details of all the decay components along with an experimental description are provided in Table S3 and Figures S21 and S22. We observe that in 10 wt % **1** in PMMA the Φ_{PL} increases from 0.447 to 0.557 when going from air to degassed conditions, while the emission lifetime increases from 2.54 to 6.14 μs . In 10 wt % **2** in PMMA, Φ_{PL} increases from 0.730 to 0.900 while the emission lifetime increases from 1.08 to 6.34 μs . The increase in lifetimes and PLQY upon removal of oxygen allows us to assign emission in films of **1** and **2** to TADF. The absence of complete quenching of long-lived emission in the films is likely due to the fact that both in neat films and when doped in PMMA, some of the chromophores will be partially encapsulated, hampering the ability of oxygen to diffuse into the film.

ELECTROLUMINESCENT DEVICES: LEECS

LEECs were prepared on top of a patterned indium tin oxide (ITO)-coated glass substrate. Prior to the deposition of the emitting layer, 80 nm of PEDOT:PSS was coated to increase the reproducibility of the cells. The emitting layer (100 nm) was prepared by spin-coating of an MeCN solution consisting of the emitting compound alone or with the addition of an ionic liquid (IL), 1-butyl-3-methylimidazolium hexafluorophosphate ([Bmim][PF₆]), at a molar ratio of 4:1 that is typically used in iTMC LEECs. To complete the devices, a layer of 70 nm of aluminum that serves as the top electrode was thermally evaporated onto the devices in a high-vacuum chamber integrated into the inert atmosphere glovebox.

To determine the performance of the LEECs, the devices were operated using a block-wave pulsed current driving method (1000 Hz and 50% of duty cycle) at different average current densities of 10, 25, and 50 A m⁻². This operational mode was selected over constant voltage mode as it decreases the turn-on time and leads to a more sustained behavior versus time.^{32,33} Using this method but also using constant-voltage driving, we were unable to observe light emission from the device employing **1** as the emitting component in single-layer devices. The driving voltage did decrease upon application of the pulsed current, indicating that electronic charges were able to inject and move through the film. However, the reason that no light emission was observed remains unclear at this moment. When **2** was employed as the emitting layer, the situation changed drastically as with these devices we did observe light emission. The luminance and average voltage are depicted in Figure 2 for LEECs using **2** (L1) and 2:IL (L2) (4:1) as the component(s) for the light-emitting layer. The figures of merit

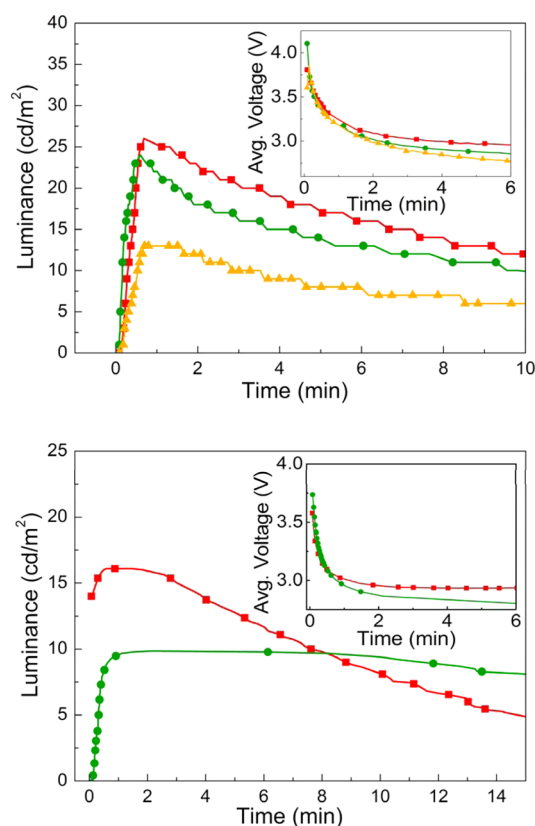


Figure 2. Luminance and average voltage (inset) for LEECs using **2** as the emissive material measured at different average current densities [50 A m⁻² (red squares), 25 A m⁻² (green circles), and 10 A m⁻² (orange triangles)]. Top device L1: ITO/PEDOT:PSS/2/Al. Bottom device L2: ITO/PEDOT:PSS/2:[Bmim][PF₆] 4:1/Al.

for the different devices studied are summarized in Table 3. Initially, for both LEECs, the average voltage applied drops

Table 3. Performance of LEEC Devices

LEEC device ^a	electroluminescence			
	average current density (A m ⁻²)	Lum _{max} (cd m ⁻²) ^a	PCE _{max} (lm W ⁻¹)	EQE _{max} (%)
L1	10	13	0.7	0.39
	25	24	0.4	0.29
	50	26	0.2	0.16
L2	25	10	0.2	0.12
	50	17	0.1	0.10

^aDevice L1: ITO/PEDOT:PSS/2/Al. Device L2: ITO/PEDOT:PSS/2:[Bmim][PF₆] 4:1/Al biased with a block wave-pulsed current at a frequency of 1000 Hz and a duty cycle of 50% at different current densities.

rapidly over the first several seconds [as also observed for the device employing **1** (not shown)]. Coinciding with the decrease in driving voltage, an increase in luminance was observed. This reached a maximal value and then slowly decreased over time. This is typical for LEECs, as the injection barrier for electrons and holes is reduced because of the migration of ions to the electrode interfaces and the subsequent formation of doped regions.^{34–36} The reduction of the injection barrier leads to the reduced driving voltage to sustain the set current density. With an increasing operating time, the doped regions expand, leading to a slowly increasing quenching of the

excitons and, as a result, a (nonpermanent) luminance reduction.³⁷

We found that the LEECs containing additional ions in the emitting layer have a luminance (and efficiency) lower than that of the LEECs with only the emitting material 2 (Figure 2). This was initially surprising as generally an increased luminance is observed upon the addition of IL to the emitting layer in LEECs employing iTMCs.^{38,39} This increased luminance is accompanied by an increase in Φ_{PL} of the emitting layers due to an increase in the interemitter distance, which leads to a reduction in the level of concentration quenching. The results, here obtained, indicate that concentration quenching effects are less important in the TADF materials than they are in iTMC-based LEECs. The addition of the IL, similar to what was observed in MeCN solution, generates a more polar environment, widening the singlet–triplet energy gap and arresting the TADF mechanism in the device.

In Figure 2, it is also clearly seen that the luminance decreases with decreasing current density, but this relationship is far from linear. A similar behavior has been observed for iTMC-based LEECs.^{40–42} This effect was attributed to a reduced level of quenching of the excitons due to either a reduction of charge carriers, excited states, or both. As the devices are operated at a fixed average current density, the efficiency of the devices is directly proportional to the luminance. With a 5-fold decrease in current density, the luminance of the 2 only (no IL, device L1) device drops only from approximately 26 to 13 cd m^{−2}. Hence, the device power efficiency strongly increases to 0.7 lm W^{−1} with an external quantum efficiency (EQE) of 0.39% (assuming Lambertian emission).

The electroluminescence (EL) spectra are shown in Figure S21. All cells emit homogeneously from the active area. The spectra are rather similar to those obtained by photoexcitation, featuring an unstructured green emission centered at 538 nm (CIE coordinates of 0.35, 0.57).

■ ELECTROLUMINESCENT DEVICES: OLEDs

We also explored 1 and 2 as the emitter layer (EML, ~20 nm) in solution-processed multilayer architecture OLEDs shown in Figure 3, in which PEDOT:PSS (~30 nm) acts as the hole

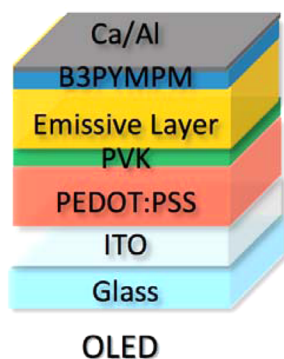


Figure 3. Schematic diagram of the device structure: The emissive layer is either pristine 1/2 or doped as mCP:OXD-7:1/2 (70:20:10). Abbreviations: PEDOT:PSS, poly(3,4-ethylenedioxythiophene):poly(styrenesulfonate); PVK, poly(*N*-vinylcarbazole); B3PYMPM, bis[4,6-(3,5-di-3-pyridylphenyl)-2-methylpyrimidine]; mCP, 3,5'-*N,N'*-dicarbazole-benzene; OXD-7, 1,3-bis[(4-*tert*-butylphenyl)-1,3,4-oxadiazolyl]phenylene.

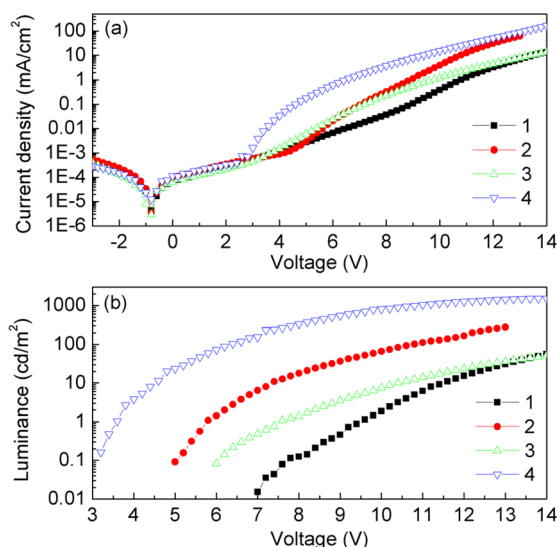


Figure 4. (a) Current density and (b) luminance vs voltage characteristics of the devices.

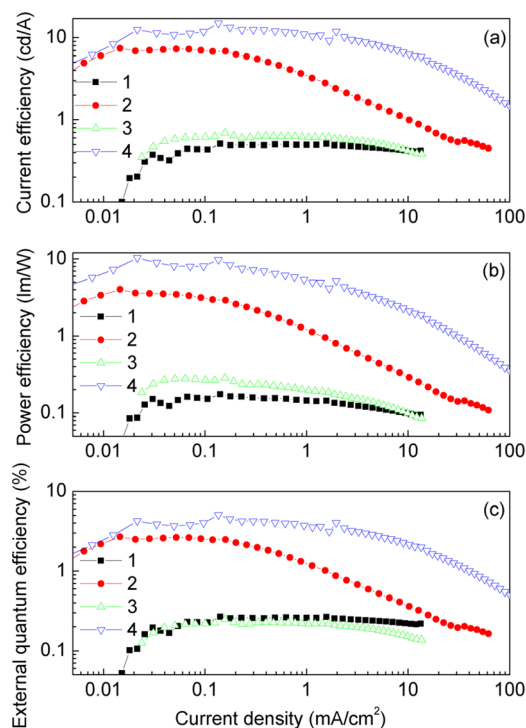


Figure 5. Current density vs current efficiency (a), power efficiency (b), and external quantum efficiency (c) curves of the four devices.

injection layer (HIL) and PVK (~30 nm) behaves as the hole-transporting layer (HTL) and electron/exciton blocking layer (EBL) because of its high lowest unoccupied molecular orbital (LUMO) level (2.2 eV) and triplet level (2.5–2.9 eV).^{43,44} The PVK was coated with the emissive layer and B3PYMPM deposited as the electron-transporting layer (ETL) because of its high electron mobility and excellent hole blocking ability⁴⁵ associated with its deep highest occupied molecular orbital (HOMO) level (6.8 eV). Calcium coated with aluminum was used as the cathode. For devices 1 and 2, the emissive layer (EML) consisted of 1 and 2 doped into mCP ($T_1 = 2.9$ eV) and OXD-7 ($T_1 = 2.7$ eV), respectively, which are cohosts for the emitters.⁴⁶ Generally, the multilayer structure can either

Table 4. Comparison of the Performance of the Four OLEDs

OLED device ^a (emitter)	V_{on}^b (V)	λ_{peak}^c (nm)	fwhm ^d (nm)	$\text{EQE}_{\text{max}}^e$ (%)	CE_{max}^f (cd/A)	PE_{max}^g (lm/W)	CIE^h
1 (doped -1)	9.5	467	148	0.3	0.5	0.2	(0.25, 0.31)
2 (doped -2)	5.8	566	137	2.7	7.4	4.0	(0.42, 0.49)
3 (neat -1)	7.5	532	132	0.3	0.7	0.3	(0.38, 0.52)
4 (neat -2)	3.6	546	128	5.1	14.9	10.3	(0.41, 0.53)

^aFor device architecture, see Figure 3 and the text. ^bTurn-on voltage at 1 cd/m². ^cPeak wavelength at 1 mA/cm². ^dFull width at half-maximum of the EL spectrum at 1 mA/cm². ^eMaximal external quantum efficiency. ^fMaximal current efficiency. ^gMaximal power efficiency. ^hThe Commission Internationale de L'Eclairage coordinates at 1 mA/cm².

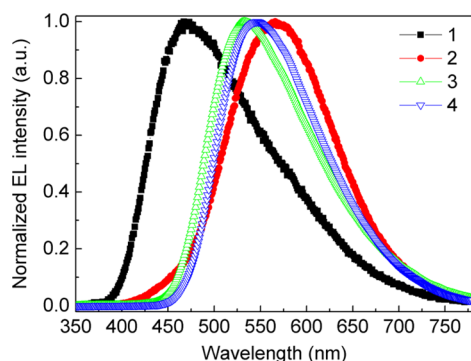


Figure 6. Comparison of the normalized EL spectra at 1 mA/cm² of the devices.

balance hole/electron in the EML or confine the excitons inside the EML to enhance the radiative recombination rate. As we found that neat 2 produced a viable LEEC, we also made devices with neat films of the emitters (1 or 2, 10 nm thick, device 3 or 4, respectively).

Figure 4a displays the current density–voltage characteristic curves of the four devices based on emitters 1 and 2. With the similar architectures, the devices exhibited similar current densities in the region of [−3 V, 3 V] where no emission was

observed. For the doped EML, device 2 showed a higher current density above 5 V and also a higher luminance. Device 2 also showed a current efficiency (CE), a power efficiency (PE), and an external quantum efficiency (EQE) much higher than those of device 1 with emitter 1 (see Figure 5). From Figure 6, we observe that the peak wavelength of device 2 was red-shifted by 99 nm compared to that of device 1. Thus, the Commission Internationale de L'Eclairage (CIE) coordinates were changed from (0.42, 0.49) to (0.25, 0.31). Surprisingly, but analogous to what was observed with the LEECs, both the nondoped devices 3 and 4 exhibited better performance (Table 4) compared to that of doped devices 1 and 2, respectively. For the devices with emitter 2, lower driving voltages were realized, especially for device 4, which required only 3.6 V to turn on. It is worth mentioning that the maximal CE, PE, and EQE were approximately double those of device 2. In contrast, devices 1 and 3 achieved similar luminous efficiencies, even though the EL spectra are significantly different (shown in Figures 6 and 7a,c). Unlike the other devices, device 2 showed current density-dependent EL spectra as the component from the cohost increased with the current density mainly due to incomplete host–guest energy transfer efficiency and exciplex emission.⁴⁷ Devices 2 and 4 showed not only better charge injection/transport but also higher luminance. Given that emitters 1 and 2 have rather similar HOMO and LUMO

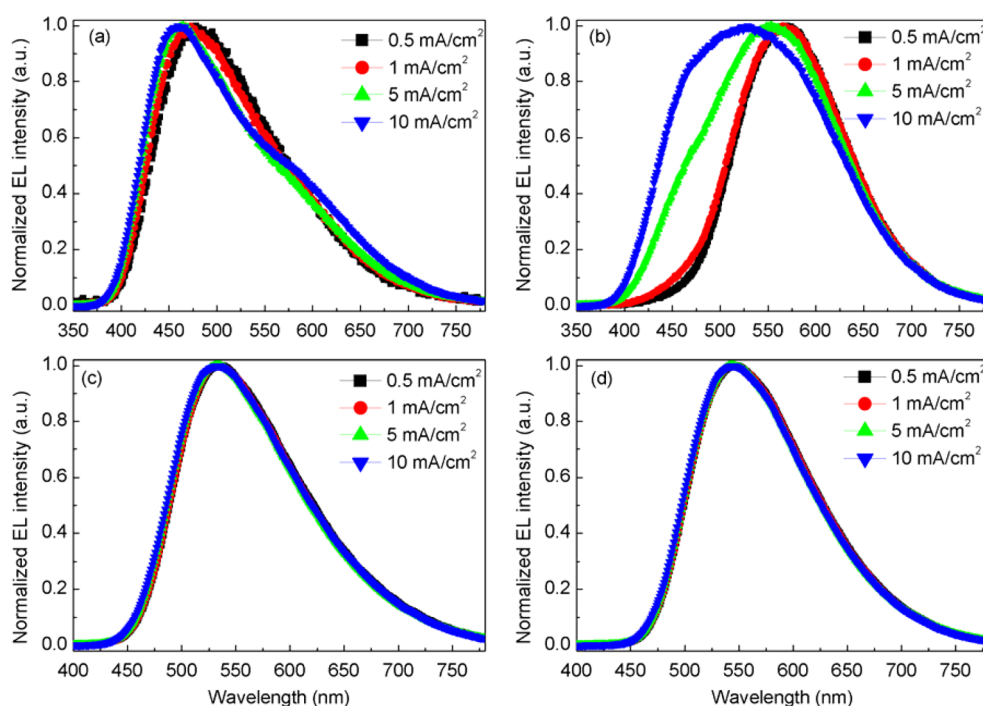


Figure 7. Normalized EL spectra as a function of the current density of (a) device 1, (b) device 2, (c) device 3, and (d) device 4.

energy levels, which are likely not the origin of the difference in charge injection/transport, it is proposed that the C_2 symmetry of emitter **2** may induce better packing in the solid state, giving better charge transport. The higher luminance is attributed to the enhanced film photoluminescence quantum yield observed in emitter **2**. From the maximal EQE of 5.1% of device **4**, we can estimate that the delayed fluorescent component of the quantum yield of **2** is ~18.4% according to eq 2,⁴⁸ which is consistent with the value of 21.4% (Table 2) calculated from the transient measurement.

$$\text{EQE} = \eta_{\text{oc}} \left[\frac{1}{4} \eta_{\text{PL}} + \frac{3}{4} \frac{\Phi_{\text{DF}}}{1 - (\eta_{\text{PL}} - \Phi_{\text{DF}})} \right] \quad (2)$$

where γ is the ratio of hole–electron recombination rate and considered to be 1 here, η_{oc} is the outcoupling efficiency (~20%), η_{PL} is the quantum yield, and Φ_{DF} is the delayed component of the quantum yield.

CONCLUSIONS

In summary, we synthesized in good yield and characterized the first charged organic TADF emitters. Using **2**, green light-emitting devices that operate as LEECs and OLEDs were prepared. For the LEECs, the operating voltages are very low with an average of 2.7 V. The efficiency increases strongly with a decreasing carrier density and reaches a maximal external quantum efficiency of ~0.4%.

We demonstrated that the solution-processed and color-stable OLEDs with pristine films outperformed the devices with the doped EML, which is rarely realized in the structure with TADF characteristics (EQE up to 5.1%). Device **4** exhibited the highest EQE of any single-component TADF OLED to date. Further optimization of the molecular structure for high quantum yield will pave the way for orthogonally spin-coated OLEDs with high luminous efficiency, which would lead to all solution-processed devices and practical application.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acs.chemmater.5b03245.

Experimental section, nuclear magnetic resonance spectra for all compounds, supplementary electrochemical data for **1** and **2**, supplementary absorption data for **1** and **2**, supplementary time-resolved luminescence data for **1** and **2**, and supplementary power intensity as a function of excitation power data for **1** and **2** (PDF)

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Notes

The authors declare no competing financial interest.

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